

# Cu-Fe-Ni-S (Copper-Iron-Nickel-Sulfur)

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The work on this important quaternary system by Craig and Kullerud and other earlier studies do not appear to have been reviewed. Three doctoral theses around 1980 [1976Lee], [1983Chu], and [1984Din], mainly on the thermodynamic properties of this system and the related subsystems, are not available in published form. This review presents the schematic phase relationships outlined by Craig and Kullerud, with additional results from more recent studies.

## Binary Systems

For brief descriptions of the Cu-Fe, Cu-S, and Fe-S systems, see the Cu-Fe-S update in this issue. Copper and nickel [both face-centered-cubic (fcc)] form a continuous solid solution over a wide range of temperature [Massalski2]. A miscibility gap arises below 354.5 °C, where the fcc phase splits into Cu-rich and Ni-rich phases. The Fe-Ni phase diagram [1991Swa] is characterized by a very narrow solidification range with a peritectic reaction at 1514 °C, between body-centered cubic (bcc)  $\delta$  and liquid that yields the Fe-based fcc solid solution. A continuous solution denoted  $\gamma$  between fcc Fe and Ni is stable over a wide range of temperature. An ordered phase  $\text{FeNi}_3$  forms congruently

from  $\gamma$  at 517 °C. The Ni-S phase diagram [Massalski2] has a number of intermediate phases:  $\text{NiS}_2$  (pyrite type cubic, mineral name vaesite denoted *vs*),  $\text{Ni}_3\text{S}_4$  ( $\text{Co}_3\text{S}_4$  type cubic, mineral name violarite, *vio*),  $\text{Ni}_{1-x}\text{S}$  (NiAs type hexagonal) and its low-temperature form called millerite (*B13* type rhombohedral),  $\text{Ni}_7\text{S}_6$  (monoclinic) and its low-temperature form godlevskite, *gd* (end-centered orthorhombic), and  $\text{Ni}_3\text{S}_2$  (fcc) and its low-temperature form heazlewoodite, *hz* (rhombohedral). The high-temperature form of  $\text{Ni}_3\text{S}_2$  has a homogeneity range.

## Ternary Systems

The review of the Cu-Fe-Ni system by [1990Gup] presented a liquidus projection, isothermal sections at 1250, 1150, 850, 750, 400, and 20 °C, the miscibility gap in the fcc ( $\gamma$ ) phase between 1050 and 600 °C, and vertical sections at 50Cu-50Fe and 90Cu-10Fe (wt.%). A recent report [2000Qin] presents four isothermal sections at 600, 800, 1000, and 1050 °C. The Cu-Fe-S system reviewed by [1979Cha1] is updated in this issue. The Cu-Ni-S system reviewed by [1979Cha2] presented the liquid miscibility gap at 1200 °C, four isothermal sections at 780, 600, 500,

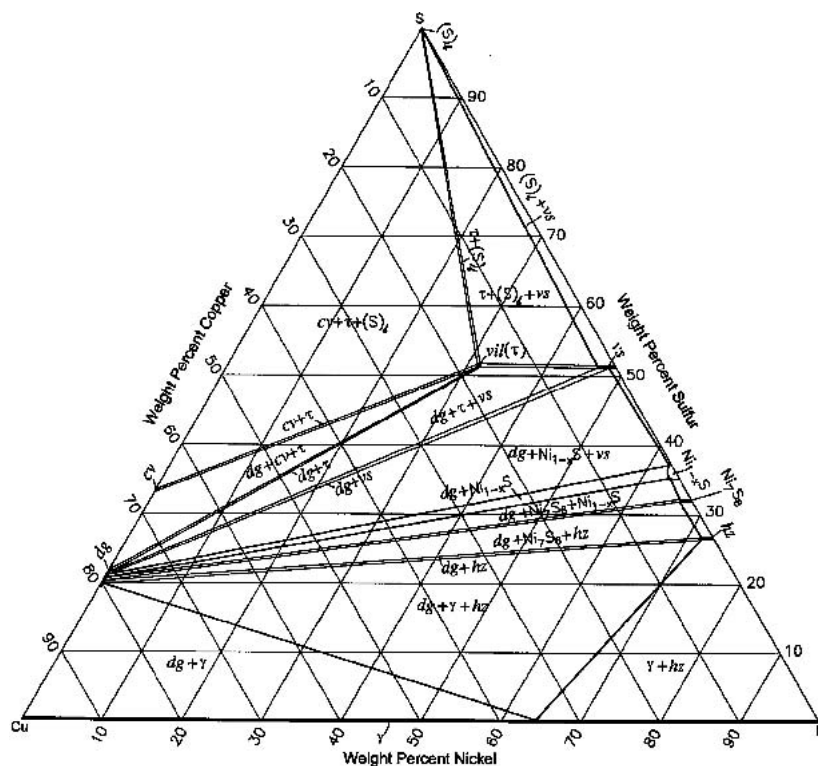
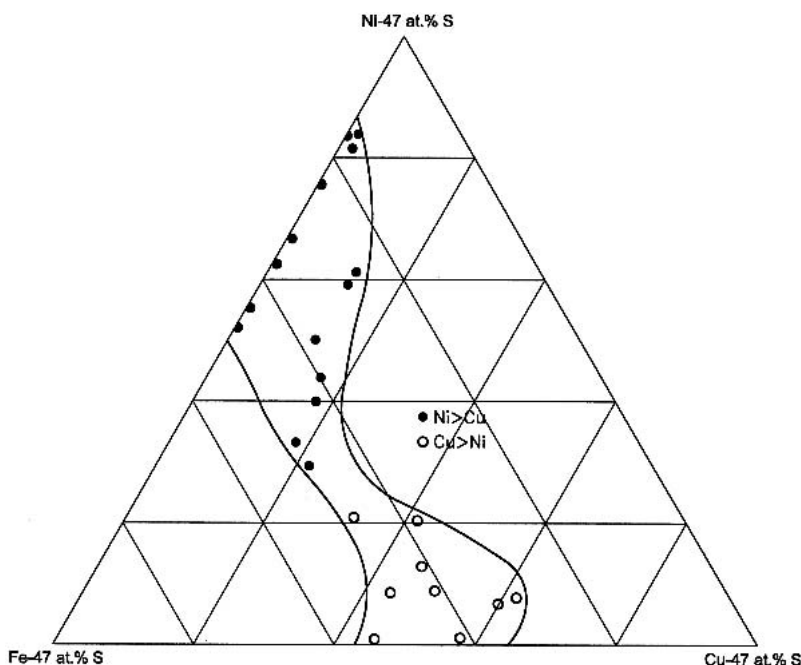


Fig. 1 Cu-Ni-S isothermal section at 500 °C [1979Cha]



## Section II: Phase Diagram Evaluations



**Fig. 5** Cu-Fe-Ni-S the continuous solid solution between *iss* and  $(\text{Ni,Fe})_3\text{S}_2$  at a constant S content of 47 at.% and at 760 °C [1995Per]

Among the other reports on this quaternary system, [1966Pop] presented a liquidus projection for the  $\text{Cu}_2\text{S}$ - $\text{FeS}$ - $\text{NiS}$  pseudoternary system. [1971Cab] measured the composition of both natural and synthetic Ni-bearing talnakhite, *tk* ( $\text{Cu}_9\text{Fe}_5\text{S}_{16}$ ) and found that up to ~0.6 at.% (0.8 wt.%) Ni substitutes for Fe in this compound. [1995Per] investigated alloy compositions that lie on the section with a constant S content of 47 at.% and found that at 760 °C, there exists a continuous quaternary solid solution between the intermediate solid solution (*iss*) of the Cu-Fe-S ternary system and the solid solution based on the high-temperature form of  $\text{Ni}_3\text{S}_2$  of the Fe-Ni-S system, which is known to dissolve large amounts of Fe (up to 33 at.% at 725 °C). These two ternary solutions have the same cubic symmetry and similar lattice parameters. Figure 5 shows this solid solution field at 760 °C. Compositions with Ni > Cu are notionally called  $(\text{Ni,Fe})_3\text{S}_2$  solid solution and those with Cu > Ni can be described as *iss*. The caption given to Fig. 5 by [1995Per] lists a constant S content of 45 at.%, whereas all the investigated compositions listed in several tables by [1995Per] give S = 47.06 at.%, corresponding to “ $(\text{Cu,Fe,Ni})_9\text{S}_8$ .” The sulfur fugacity of the solid solution measured by [1995Per] by the pyrrhotite method gave  $\log f_{\text{S}_2}(\text{atm}) = -13.8$  at the most Fe-rich composition. This increases progressively with increasing Ni in the solution and further on with increasing Cu to a maximum value of -7.

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